

Electromagnetic Radiation Driven Phase Transition in Silver Telluride-Iron Oxide and Iron Telluride Nano-Composites

Yang Bao^{1,3}, Wei Zheng², Praveen Gurralla³, Biao Xu⁴, Jiming Song³, and Yue Wu²

¹ College of Electronic and Optical Engineering
Nanjing University of Posts and Telecommunications, Nanjing, Jiangsu 210023, China
brianbao@njupt.edu.cn

² Department of Chemical and Biological Engineering
Iowa State University, Ames, IA 50011, USA
zhengwei@iastate.edu; yuewu@iastate.edu

³ Department of Electrical and Computer Engineering
Iowa State University, Ames, IA 50011, USA
praveeng@iastate.edu; jisong@iastate.edu

⁴ School of Chemical Engineering
Nanjing University of Science and Technology, Nanjing, Jiangsu 210094, China
xubiao@njust.edu.cn

Abstract — Temperature-dependent switchability of electrical property has drawn a lot of attention recently. In this article, electromagnetic (EM) radiation is used to induce temperature change in silver telluride and iron telluride nanostructures, which can be potentially used to trigger electrical property changes in them. Due to the low EM absorptivity of silver telluride nanowire, heating it via EM fields is very challenging. To enhance its EM absorptivity, various percentages (by mass) of iron oxide powder are mixed into it. It is verified that, similar to pure silver telluride, the mixture also exhibits a rapid change in electrical conductivity within a certain temperature range. The experiment is designed to characterize the EM absorptivity of three such mixtures with different percentage compositions and an iron telluride nanodisk in the X-band frequency range (8 to 12 GHz). The temperature increase induced by the absorbed EM energy will lead to the rapid change of electrical conductivity by α -type to β -type phase transition in silver telluride and p-type to n-type transition in iron telluride, which can be potentially used to develop EM sensors for applications in communication technology.

Index Terms — Electrical switchability, electromagnetic radiation absorption, EM sensor, nano-composites.

I. INTRODUCTION

With the rapid development of wireless communication, the increasing usage of electromagnetic devices results in serious electromagnetic interference

(EMI) and electromagnetic compatibility (EMC) problems [1-4]. More seriously, the high density of electromagnetic (EM) radiation caused by wireless communication has been confirmed to have a great effect on the health and safety of humans, such as sleep disturbance, headache, nausea, visual disorders, respiratory problems, and nervous excitation [5].

In consideration of the pollution and threat from EM radiation, many ideas have been proposed to prevent the radiation from being harmful to the humans and communities. EM sensors or switches have since come out. There are some sensors or switches based on amorphous semiconductor [6], superconductor [7], and high-pressure gas arresters [8]. However, semiconductor-based switches have a limited number of operations when protecting circuits against high voltage transients, superconductors require low-temperature cooling, and gas arresters have complicated design with relatively large size [9]. After that, in recent years, significant progress has been achieved in microwave cryogenic electronics operating at liquid nitrogen temperatures [10]. Symmetrical thin-film switches based on current-induced phase transitions from superconductor to normal state [11], and polycrystalline manganite films exhibiting the “electro-resistance” effect [12] were proposed. However, the high speed of operation of these sensors is always accompanied with energy focusing in narrow channels of the film and damage to the protector [10]. To overcome the drawbacks mentioned above, the potential use of silver telluride and iron telluride

nanostructures in EM sensors are considered, where Joule heating induced by the EM energy absorbed by these materials triggers rapid electrical property changes.

In this article, silver telluride nanowire, which has low EM absorptivity, is blended with iron oxide powder, which has higher ability to absorb EM radiation energy and subsequently convert it into heat or other forms of internal energy [13-17], to improve the mixtures' EM absorptivity while keeping the feature of rapid electrical property changes [18-26]. The EM absorption properties of three mixture samples with different percentages of silver telluride nanowire and iron oxide powder, and a nanodisk made of iron telluride are measured in X band, which is in great demand [27-30]. The good EM absorption properties of these materials can lead to increase in their temperature in environments with high EM radiation. This will eventually cause the rapid change of the electrical conductivity by α -type to β -type phase transition [31-32] in silver telluride and p-type to n-type transition [33-34] in iron telluride, which can be potentially applied to develop EM sensors. The remainder of this paper is organized as follows. Section II describes the experimental details. Section III shows the results and discussion. Finally, the conclusion is drawn in Section IV.

II. EXPERIMENTAL DETAILS

All chemicals are used as received without further purification. Tellurium dioxide ($\geq 99.99\%$), polyvinylpyrrolidone (PVP, MW~ 40,000), potassium hydroxide, (KOH, $\geq 90\%$), hydrazine monohydrate ($\text{N}_2\text{H}_4\cdot\text{H}_2\text{O}$, 78%~82%), and iron oxide (Fe_3O_4 , $\geq 99.99\%$, 325 mesh) were purchased from Sigma Aldrich, while silver nitrate (AgNO_3 , $\geq 99.9\%$), ethylene glycol (EG, $\geq 99\%$) and ethanol ($\geq 95\%$) were purchased from VWR.

For the synthesis of $\text{Ag}_2\text{Te}_{2x}\text{Ag}$ nanowires, the procedure from our previous publication was followed exactly [32]. Firstly, 9.576 g TeO_2 , 12 g PVP, 44.484 g KOH, and 600 ml EG were added to a 1 L glass reactor with magnetic stirring initiated for continuous mixing. The glass reactor was heated to 120°C and 20 ml $\text{N}_2\text{H}_4\cdot\text{H}_2\text{O}$ was rapidly injected into the reactor. Here the TeO_2 served as a precursor for Tellurium, $\text{N}_2\text{H}_4\cdot\text{H}_2\text{O}$ was a reductant to provide electrons with Te, while KOH was added to tune the pH of the solution and the reaction rate. The temperature was maintained at 120°C for 1 hour under nitrogen gas protection. Then, the reaction was stopped and cooled down to room temperature naturally. The tellurium nanowire was washed three times with deionized (DI) water and re-dispersed in 800 ml EG in a 2 L beaker for Ag_2Te synthesis. Alongside, 40.769 g AgNO_3 was dissolved in 200 ml EG. Then, the AgNO_3/EG solution was added into the 2 L beaker and stirred at room temperature for 2 hours for the conversion from Te into Ag_2Te . In this process, AgNO_3 was reduced to metallic Ag and then combined with

Te to yield Ag_2Te . The as-synthesized $\text{Ag}_2\text{Te}_{2x}\text{Ag}$ nanowires were centrifuged two times with deionized water and washed in an ethanol solution (12.5% of $\text{N}_2\text{H}_4\cdot\text{H}_2\text{O}$) for 24 hours to remove the surfactant. The mixture was then washed twice with ethanol and vacuum-dried.

$\text{Ag}_2\text{Te}-\text{Fe}_3\text{O}_4$ composite sintering: The Ag_2Te nanowires was grounded into loose powder and gridded by a 325-mesh sieve in a nitrogen-filled glovebox. Ag_2Te and Fe_3O_4 powder were mixed evenly by mortar and loaded into graphite die for spark plasma sintering (SPS). The sintering condition was under 40 MPa at 850°C for 5 minutes and then cooled down to room temperature naturally [33].

The EM absorption property was measured using a programmable vector network analyzer (VNA, Agilent E8364) with calibration kit (Agilent 85052D) and cables (GORE Microwave). The samples were placed inside an X band waveguide (Aircom Microwave) to measure the S parameters.

III. RESULTS AND DISCUSSION

The Ag_2Te is analyzed by using X-ray diffraction (XRD) and transmission electron microscopy (TEM). The XRD result, Fig. 1 (a), shows that our synthesized Ag_2Te can be indexed as pure monoclinic phase silver telluride (red lines: JCPDS # 34-1042). Figure 1 (b) is the low-magnification TEM image of Ag_2Te and Fig. 1 (c) is the high-resolution TEM (HRTEM) image of Ag_2Te with its fast Fourier transform (FFT) image. The synthesized Ag_2Te is further proved to be the same monoclinic phase of Ag_2Te (space group is 13).

This large-scale solution-synthesis method enables us to sinter several $\text{Ag}_2\text{Te}-\text{Fe}_3\text{O}_4$ composite disks with 1 cm diameter by Spark Plasma Sintering (SPS) and investigate the electrical properties of the resulting mixtures. Figure 2 (a) is the XRD results of $\text{Ag}_2\text{Te}-\text{Fe}_3\text{O}_4$ composites, where all percentages are by weight. There are both silver telluride peaks (JDCPS #34-1042) and iron oxide peaks (JCPDS #19-0692) in all three composite disks. With increasing Ag_2Te weight percent, Ag_2Te peaks increase while Fe_3O_4 peaks decrease. The Seebeck coefficient of $\text{Ag}_2\text{Te}-\text{Fe}_3\text{O}_4$ composites has been measured from 40°C to 220°C and is shown in Fig. 2 (b). All these three samples have negative Seebeck coefficient values during the test temperature range which indicate they are n-type semiconductors.

For 25% $\text{Ag}_2\text{Te}-75\% \text{Fe}_3\text{O}_4$ and 50% $\text{Ag}_2\text{Te}-50\% \text{Fe}_3\text{O}_4$ samples, the absolute Seebeck coefficient increases with temperature. However, the absolute Seebeck coefficient of 75% $\text{Ag}_2\text{Te}-25\% \text{Fe}_3\text{O}_4$ decreases first then increases after 125°C . Figure 2 (c) is the electrical conductivity of $\text{Ag}_2\text{Te}-\text{Fe}_3\text{O}_4$ composites. The 25% $\text{Ag}_2\text{Te}-75\% \text{Fe}_3\text{O}_4$ sample has the highest electrical conductivity with the value between 140 S/cm to 160 S/cm. The electrical conductivity of all three samples

have a dramatic decrease between 140°C to 160°C.

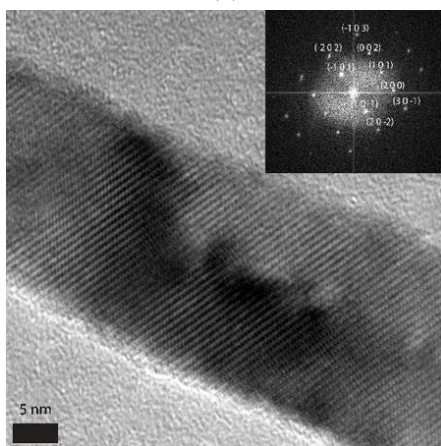
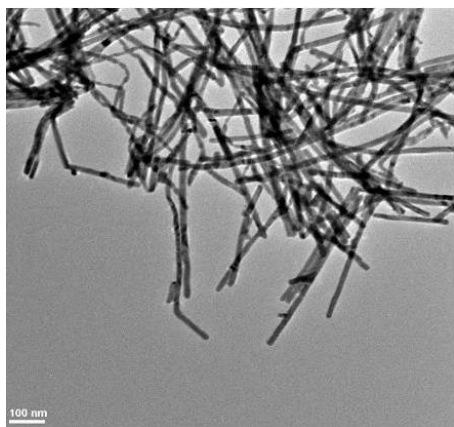
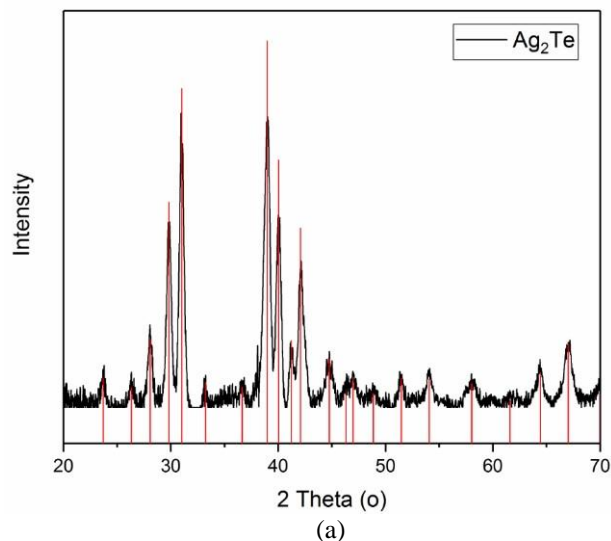


Fig. 1. (a) XRD of Ag_2Te . Red lines: standard silver telluride. (b) TEM image of Ag_2Te . (c) HRTEM of Ag_2Te (Inset: FFT of Ag_2Te).

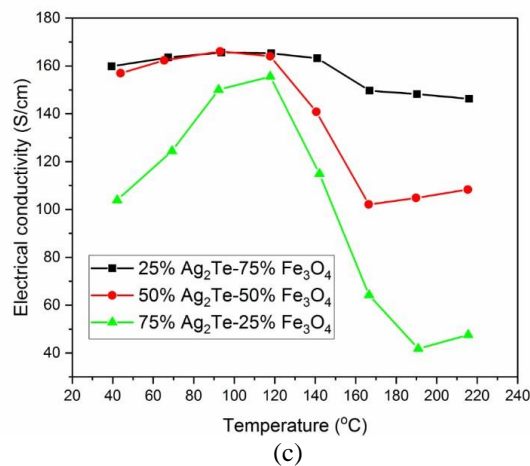
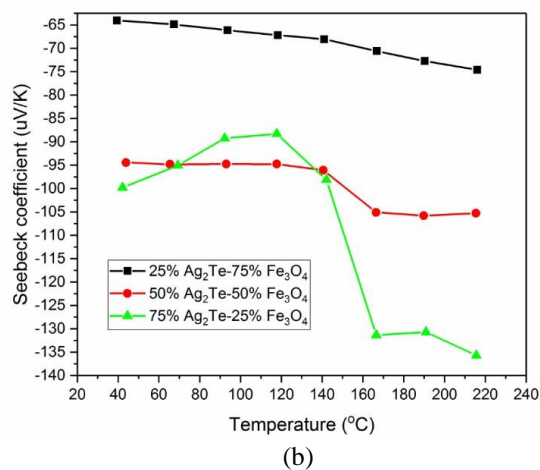
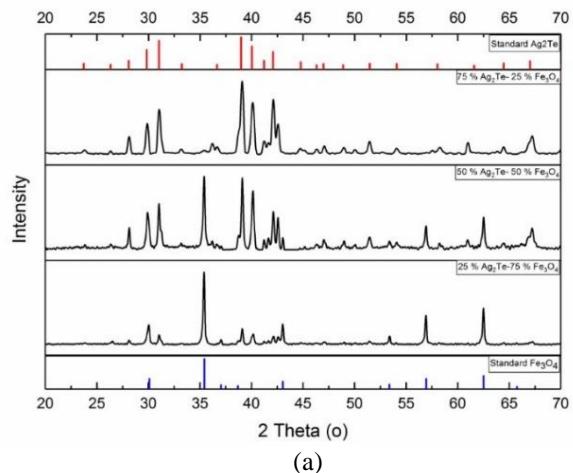


Fig. 2. (a) XRD results of $\text{Ag}_2\text{Te} - \text{Fe}_3\text{O}_4$ composites. All these percent numbers are weight percentages. (b) Seebeck coefficient of $\text{Ag}_2\text{Te} - \text{Fe}_3\text{O}_4$ composites. (c) Electrical conductivity of $\text{Ag}_2\text{Te} - \text{Fe}_3\text{O}_4$ composites.

As is well known, Ag_2Te changes from low-temperature monoclinic phase (α phase) to high-temperature cubic phase (β phase) at 150°C [35-36].

During phase transition from α phase to β phase, the energy band gap of Ag_2Te increases, which in turn decreases the carrier concentration (n) while the electron mobility (v) fluctuates a little bit. As a result, the electrical conductivity changes from a highly conductive state (α phase) to a poorly conductive state (β phase) based on the equation $\sigma = nev$ [35, 37, 38]. This phase transition property may explain the dramatic changes in Figure 2b and 2c, especially for the sample of 75% Ag_2Te -25% Fe_3O_4 . As to the FeTe_2 , it shows a temperature-dependent reversible and reproducible switching behavior between p-type to n-type conduction [33].

Magnetite is one of the most important resources for production of iron via microwave heating because of its good coupling with microwaves and three mechanisms including Joule loss, dielectric loss, and magnetic loss would contribute to microwaves heating [39]. Thus, the magnetite is mixed with the silver telluride nanowire to increase the EM absorption of the composites. Figure 3 shows the EM absorption of the three $\text{Ag}_2\text{Te} - \text{Fe}_3\text{O}_4$ composite disks with different mass percentage compositions (75% Fe_3O_4 -25% Ag_2Te , 50% Fe_3O_4 -50% Ag_2Te , 25% Fe_3O_4 -75% Ag_2Te), and the iron telluride nanodisk in X band. The measurement was performed by placing the disks in an X-band waveguide with the disk face parallel to the waveguide cross-section to ensure the same experiment condition. The input power p_0 is 0 dBm. All disks are 1.6 mm thick. The EM absorption is calculated from the measured S parameters and is given by $(1 - |S_{11}|^2 - |S_{21}|^2) p_0$ [40-42].

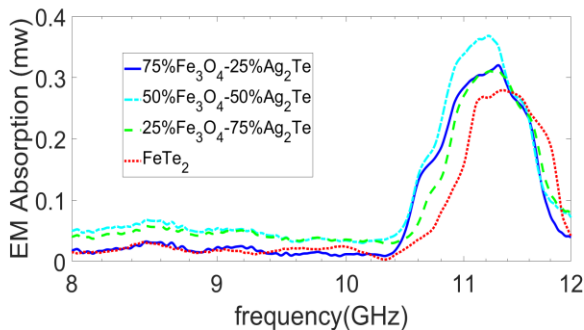


Fig. 3. EM absorption of samples in X band.

The maximum EM absorption power measured is 0.37 mW, which occurs at 11.2 GHz with the composite made of 50% Fe_3O_4 -50% Ag_2Te . This composite also shows better EM absorption performance over others in the frequency range from 8 GHz to 11.4 GHz. For the composites made of 25% Fe_3O_4 -75% Ag_2Te , 75% Fe_3O_4 -25% Ag_2Te , and the iron telluride, the maximum absorption power is 0.31 mW at 11.31 GHz, 0.32 mW at

11.29 GHz and 0.28 mW at 11.35 GHz, respectively. At least 30% power is absorbed by the samples at around 11 GHz. The potential sensors can absorb high-power EM radiation and convert it to heat, which results in a temperature rise given by $\Delta T = Q/C$, where Q is the amount of the heat absorbed and C is the heat capacity [43]. The temperature rise eventually leads to rapid change in electrical conductivity by α -type to β -type phase transition for $\text{Ag}_2\text{Te} - \text{Fe}_3\text{O}_4$ samples as shown in Fig. 2, and p-type to n-type transition in iron telluride with reversible and reproducible switching behavior [33]. A multi-physics simulation package is needed to include electromagnetic modeling of energy absorptivity of the mixtures at different frequencies, thermal modeling of temperature changes, and chemical modeling of electrical conductivity changes from α -type to β -type phase transition in silver telluride and p-type to n-type transition in iron telluride. The electromagnetic radiation driven phase transitions in $\text{Ag}_2\text{Te} - \text{Fe}_3\text{O}_4$ and FeTe_2 nano-composites can be potentially applied to develop EM sensors.

IV. CONCLUSIONS

It is shown in this work that the iron telluride and mixture composites made of silver telluride nanowire and iron oxide powder have good EM absorption in X band and exhibit rapid changes of electrical property in a certain temperature range. The composite made of 50% Fe_3O_4 -50% Ag_2Te has the maximum EM absorption power at 11.2 GHz which leads to the temperature rise. In turn, the temperature rise causes a rapid change in the electrical conductivity by α -type to β -type phase transition in the composite mixtures and p-type to n-type transition in iron telluride, which can be potentially exploited for developing EM sensors.

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